

Rare-earth site splitting in  $\text{Sm}_3\text{MoO}_7$ N. Barrier,<sup>a</sup> P. Gall<sup>b</sup> and P. Gougeon<sup>b\*</sup><sup>a</sup>Laboratoire CRISMAT, UMR CNRS 6508, 6 Boulevard Maréchal Juin, 14050 Caen Cedex 4, France, and <sup>b</sup>Sciences Chimiques de Rennes, UMR CNRS No. 6226, Université de Rennes I, Avenue du Général Leclerc, 35042 Rennes Cedex, France  
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Trisamarium molybdenum heptaoxide,  $\text{Sm}_3\text{MoO}_7$ , is isomorphous with  $\text{Ln}_3\text{MO}_7$  ( $\text{Ln} = \text{La}$  and  $\text{Pr}$ ). The crystal structure consists of chains of corner-linked  $\text{MoO}_6$  octahedra running parallel to the  $b$  axis and separated from each other by seven- or eight-coordinate  $\text{Sm-O}$  polyhedra. In contrast to  $\text{La}_3\text{MO}_7$  and  $\text{Pr}_3\text{MO}_7$ , a splitting of one  $\text{Sm}$  site into two positions is observed.

## Comment

Metal oxides of general formula  $\text{Ln}_3\text{MO}_7$ , where  $M$  is a pentavalent  $4d$  or  $5d$  transition metal such as Nb, Mo, Ru, Ta, Re, Os or Ir, and  $\text{Ln}$  is a trivalent rare-earth metal, present an ordered double fluorite structure and crystallize in various orthorhombic space groups, such as  $Pnma$ ,  $Cmcm$ ,  $C222_1$  or  $P2_12_12_1$ . The main structural feature of the  $\text{Ln}_3\text{MO}_7$  compounds is the presence of isolated zigzag chains of *trans*-corner-linked  $\text{MO}_6$  octahedra that are separated by seven- or

eight-coordinate  $\text{Ln-O}$  polyhedra. Because of this quasi-one-dimensionality,  $\text{La}_3\text{RuO}_7$ ,  $\text{Ln}_3\text{OsO}_7$  ( $\text{Ln} = \text{La}$ ,  $\text{Pr}$ ,  $\text{Nd}$ ,  $\text{Sm}$ ,  $\text{Eu}$  and  $\text{Gd}$ ; Lam *et al.*, 2002; Plaisier *et al.*, 2002; Gemmill *et al.*, 2005),  $\text{Ln}_3\text{MoO}_7$  (Greedan *et al.*, 1997; Nishimine *et al.*, 2005),  $\text{Ln}_3\text{RuO}_7$  ( $\text{Ln} = \text{La}$ ,  $\text{Pr}$ ,  $\text{Sm}$  and  $\text{Eu}$ ; Khalifah *et al.*, 2000; Wiss *et al.*, 2000; Harada & Hinatsu, 2001; Gemmill *et al.*, 2004),  $\text{Ln}_3\text{ReO}_7$  ( $\text{Ln} = \text{Pr}$ ,  $\text{Nd}$ ,  $\text{Sm}$ ,  $\text{Gd}$ ,  $\text{Tb}$  and  $\text{Dy}$ ; Wltschek *et al.*, 1996; Lam *et al.*, 2003; Hinatsu *et al.*, 2004) and  $\text{Pr}_3\text{MO}_7$  ( $M = \text{Nb}$  and  $\text{Ta}$ ; Vente *et al.*, 1994) have been extensively studied for their physical properties since peculiar magnetic and electronic properties are expected. We present here the crystal structure of  $\text{Sm}_3\text{MoO}_7$ . This compound was first synthesized as a powder sample by Prévost-Czeskleba (1987) and found to crystallize in the orthorhombic space group  $Cmcm$  as  $\text{Nd}_3\text{NbO}_7$  (Rossel, 1979). Our investigation on a single crystal indicates that  $\text{Sm}_3\text{MoO}_7$  crystallizes in the noncentrosymmetric space group  $P2_12_12_1$  as  $\text{La}_3\text{MO}_7$  (Greedan *et al.*, 1997) and  $\text{Pr}_3\text{MO}_7$  (Barrier & Gougeon, 2003). However, a partial disorder is observed at one rare-earth site, *i.e.*  $\text{Sm}1$ , in the title compound.

Perspective views of  $\text{Sm}_3\text{MoO}_7$  along the  $b$  and  $c$  axes are shown in Figs. 1 and 2, respectively. The main structural feature of  $\text{Sm}_3\text{MoO}_7$  is the occurrence of infinite single chains of tilted corner-linked  $\text{MoO}_6$  octahedra running parallel to the  $b$  axis. These chains alternate with rows of edge-shared  $\text{Sm}1\text{O}_7$  or  $\text{Sm}1'\text{O}_8$  polyhedra to form slabs parallel to the  $ab$  plane. The slabs are separated by the  $\text{Sm}2$  and  $\text{Sm}3$  cations, which are both seven-coordinated by O atoms. The  $\text{Mo-O}$  distances within the  $\text{MoO}_6$  octahedra (Fig. 3) range from 1.851 (3) to 2.088 (3) Å [1.861 (3)–2.098 (4) and 1.852 (4)–2.088 (5) Å in  $\text{La}_3\text{MO}_7$  and  $\text{Pr}_3\text{MO}_7$ , respectively], with an average value of 1.966 Å compared with 1.981 and 1.974 Å in  $\text{La}_3\text{MO}_7$  and  $\text{Pr}_3\text{MO}_7$ , respectively. Because of the octahedral tilt, the  $\text{Mo-O5-Mo}^{\text{iv}}$  angle [symmetry code: (iv)  $-x + 1, y + \frac{1}{2}$ ,

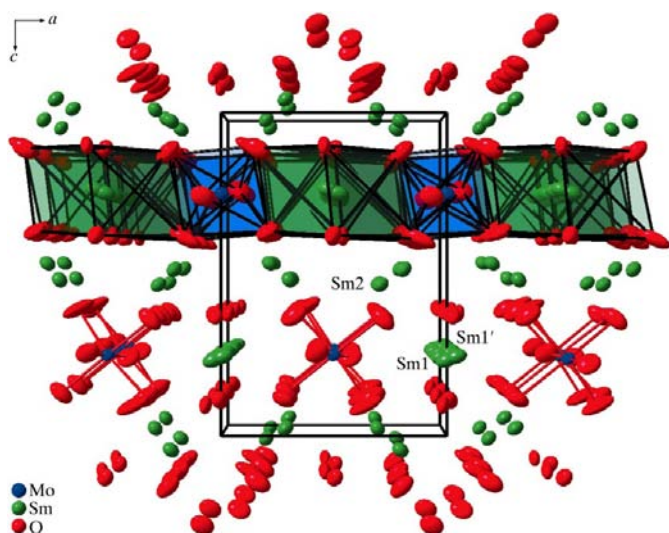


Figure 1

A perspective view of  $\text{Sm}_3\text{MoO}_7$  along the  $b$  axis. Displacement ellipsoids are drawn at the 97% probability level.

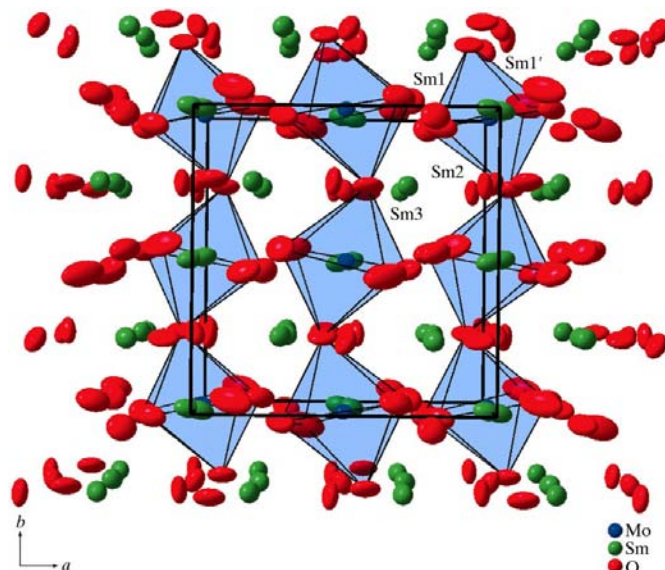
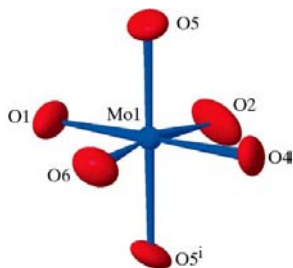


Figure 2

A perspective view of  $\text{Sm}_3\text{MoO}_7$  along the  $c$  axis. Displacement ellipsoids are drawn at the 97% probability level.

$-z + \frac{3}{2}$ ] along the chain differs significantly from  $180^\circ$ , with a value of  $146.62(14)^\circ$ . The main difference with  $\text{La}_3\text{MoO}_7$  and  $\text{Pr}_3\text{MoO}_7$  concerns the Sm1 site, which is split into two independent parts, Sm1 and Sm1'. Atom Sm1, with 92.0(3)% occupancy, has a coordination number of 7, with five short Sm—O bonds in the range 2.348(3)–2.383(3) Å and two long Sm—O bonds of 2.637(3) and 2.637(4) Å. This seven-coordination may be described as a bicapped square-based pyramid and corresponds to the rare-earth site 1 in  $\text{La}_3\text{MoO}_7$  and  $\text{Pr}_3\text{MoO}_7$ . In contrast, atom Sm1', with a site occupancy of 0.080(3), has a coordination number of 8, with six Sm—O bonds distributed in the range 2.269(6)–2.585(6) Å and two others of length 2.969(8) and 2.997(8) Å. The six nearest O atoms form a highly distorted octahedron and the other two cap two faces. Atoms Sm2 and Sm3 are in distorted pentagonal bipyramidal configurations, with Sm—O distances in the ranges 2.273(2)–2.642(2) and 2.243(3)–2.542(2) Å, respectively.

The phenomenon of rare-earth site splitting has recently been observed for the related compounds  $\text{Tb}_3\text{RuO}_7$  and  $\text{Dy}_3\text{RuO}_7$  (Ishizawa, Suwa & Tateishi, 2007; Ishizawa, Suwa, Tateishi & Hester, 2007). Indeed, these compounds, which crystallize in the noncentrosymmetric space group  $P2_1nb$  with a double  $b$  axis ( $a$  axis in  $\text{Sm}_3\text{MoO}_7$ ), present two rare-earth atom sites (out of six crystallographically independent ones) that are split into two positions. As in the ruthenium compounds, the rare-earth sites that are split in the title compound are those that alternate with the  $\text{MoO}_6$  chains. Our study shows clearly that such a partial disorder can exist with transition metals other than ruthenium in the  $\text{Ln}_3\text{MO}_7$  series. Examination of various structures of the  $\text{Ln}_3\text{MO}_7$  type shows that, in a number of them, the rare-earth metals that alternate with the  $\text{MoO}_6$  chains present either equivalent isotropic displacement parameters that are, at least, double those of the other rare-earth atoms (Wltschek *et al.*, 1996; Khalifah *et al.*, 2000; Lam *et al.*, 2003; Hinatsu *et al.*, 2004; Gemmill *et al.*, 2005) or large prolate anisotropic atomic displacement parameters (Barrier & Gougeon, 2003; Ishizawa *et al.*, 2006). Finally, one can note that the rare-earth disorder seems to occur not only in the noncentrosymmetric space groups but also in the centrosymmetric ones, and it would be interesting to see if this partial structural disorder depends on the rare-earth atom size.



**Figure 3**  
The numbering scheme of the  $\text{MoO}_6$  octahedron. Displacement ellipsoids are drawn at the 97% probability level. The symmetry code is as in Table 1.

## Experimental

Single crystals of  $\text{Sm}_3\text{MoO}_7$  were prepared from a stoichiometric amount of  $\text{Sm}_2\text{O}_3$ ,  $\text{MoO}_3$  and Mo. Before use, the Mo powder was reduced under a hydrogen flow at 1273 K for 6 h and the rare-earth oxide was pre-fired at 1273 K overnight and left at 873 K. The initial mixture (*ca* 5 g) was cold pressed and loaded into a molybdenum crucible, which was sealed under a low argon pressure using an arc welding system. The charge was heated at a rate of  $300\text{ K h}^{-1}$  to 1973 K, at which temperature it was held for 10 min, then cooled at a rate of  $100\text{ K h}^{-1}$  to 1373 K and finally furnace cooled.

### Crystal data

$\text{Sm}_3\text{MoO}_7$	$V = 600.21(2)\text{ \AA}^3$
$M_r = 658.99$	$Z = 4$
Orthorhombic, $P2_12_12_1$	Mo $K\alpha$ radiation
$a = 7.4500(1)\text{ \AA}$	$\mu = 30.94\text{ mm}^{-1}$
$b = 7.5460(1)\text{ \AA}$	$T = 293(2)\text{ K}$
$c = 10.6766(2)\text{ \AA}$	$0.19 \times 0.09 \times 0.04\text{ mm}$

### Data collection

Nonius KappaCCD diffractometer	17049 measured reflections
Absorption correction: analytical (de Meulenaer & Tompa, 1965)	3719 independent reflections
$T_{\min} = 0.037$ , $T_{\max} = 0.367$	3438 reflections with $I > 2\sigma(I)$
	$R_{\text{int}} = 0.034$

### Refinement

$R[F^2 > 2\sigma(F^2)] = 0.022$	$\Delta\rho_{\max} = 1.94\text{ e \AA}^{-3}$
$wR(F^2) = 0.042$	$\Delta\rho_{\min} = -3.05\text{ e \AA}^{-3}$
$S = 1.05$	Absolute structure: Flack (1983), 1593 Friedel pairs
3719 reflections	Flack parameter: 0.491(17)
107 parameters	
1 restraint	

**Table 1**

Selected bond lengths (Å).

Mo1—O2	1.851(3)	Sm1'—O4	2.585(6)
Mo1—O6	1.913(3)	Sm1'—O6	2.969(8)
Mo1—O5	1.939(2)	Sm1'—O2 <sup>ii</sup>	2.997(8)
Mo1—O5 <sup>i</sup>	2.000(2)	Sm2—O7 <sup>iii</sup>	2.273(2)
Mo1—O1	2.002(3)	Sm2—O7	2.313(2)
Mo1—O4	2.088(3)	Sm2—O4 <sup>iv</sup>	2.361(3)
Sm1—O7 <sup>ii</sup>	2.348(3)	Sm2—O1	2.385(3)
Sm1—O4	2.376(3)	Sm2—O2 <sup>v</sup>	2.438(3)
Sm1—O7 <sup>i</sup>	2.378(3)	Sm2—O6 <sup>vi</sup>	2.442(3)
Sm1—O3 <sup>ii</sup>	2.382(3)	Sm2—O5	2.642(2)
Sm1—O3 <sup>i</sup>	2.383(3)	Sm3—O3	2.243(3)
Sm1—O1 <sup>ii</sup>	2.637(3)	Sm3—O3 <sup>vii</sup>	2.272(3)
Sm1—O6	2.637(4)	Sm3—O4 <sup>viii</sup>	2.344(3)
Sm1'—O7 <sup>i</sup>	2.269(6)	Sm3—O1 <sup>ix</sup>	2.399(3)
Sm1'—O3 <sup>ii</sup>	2.308(6)	Sm3—O6 <sup>iv</sup>	2.465(3)
Sm1'—O1 <sup>ii</sup>	2.420(6)	Sm3—O2	2.498(3)
Sm1'—O3 <sup>i</sup>	2.448(6)	Sm3—O5	2.542(2)
Sm1'—O7 <sup>ii</sup>	2.451(6)		

Symmetry codes: (i)  $-x + 1, y - \frac{1}{2}, -z + \frac{3}{2}$ ; (ii)  $x - 1, y, z$ ; (iii)  $x - \frac{1}{2}, -y + \frac{3}{2}, -z + 1$ ; (iv)  $-x + 1, y + \frac{1}{2}, -z + \frac{3}{2}$ ; (v)  $-x + \frac{3}{2}, -y + 1, z - \frac{1}{2}$ ; (vi)  $x + \frac{1}{2}, -y + \frac{3}{2}, -z + 1$ ; (vii)  $x - \frac{1}{2}, -y + \frac{3}{2}, -z + 2$ ; (viii)  $x + \frac{1}{2}, -y + \frac{3}{2}, -z + 2$ ; (ix)  $-x + \frac{3}{2}, -y + 1, z + \frac{1}{2}$ .

Systematic absences were only consistent with the noncentrosymmetric space group  $P2_12_12_1$ . The atomic coordinates of Pr, Mo and O from  $\text{Pr}_3\text{MoO}_7$  (Barrier & Gougeon, 2003) were used as starting positions in the first stages of the refinement in the present study. The least-squares refinement with anisotropic atomic displacement parameters for all atoms yielded an  $R$  factor of 0.027 for 3438 reflections. At this stage, however, a relatively large residual electron-density peak of  $8.1\text{ e \AA}^{-3}$  was found at about  $0.58\text{ \AA}$  from atom Sm1. To model the electron density of the Sm1 site, it was necessary to split it into two independent sites, Sm1 and Sm1', separated by  $0.58\text{ \AA}$ . The split pair Sm1/Sm1' was constrained to have

the same anisotropic atomic displacement parameter in the later refinements. The site-occupation factors of atoms Sm1 and Sm1' were first refined freely, leading to a sum of 1.001 (3). Consequently, the sum was constrained to be 1 in the final cycles of refinement. This split-atom model decreased the *R* factor to 0.022 and the residual electron density near the Sm1 and Sm1' sites to 1.79 e Å<sup>-3</sup>. An attempt to refine the structure in the space group *Pnma*, as suggested by *PLATON* (Spek, 2003), was unsuccessful and led to an *R* factor of about 0.10. The Flack (1983) parameter refined to 0.491 (17), indicating that the crystal contained a mixture of the two absolute structures. The highest peak and the deepest hole in the final Fourier map are located 0.59 and 0.52 Å, respectively, from atom Sm2.

Data collection: *COLLECT* (Nonius, 1998); cell refinement: *COLLECT*; data reduction: *EVALCCD* (Duisenberg, 1998); program(s) used to refine structure: *SHELXL97* (Sheldrick, 1997); molecular graphics: *DIAMOND* (Brandenburg, 2001); software used to prepare material for publication: *SHELXL97*.

Supplementary data for this paper are available from the IUCr electronic archives (Reference: FA3110). Services for accessing these data are described at the back of the journal.

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